

Abstract: In this study we present the results of electronic structure calculations and the experimental investigations on the structural and magnetic properties of the $Mn_{50}Al_{46}M_4$ ($M = Mn, Ni, Ti$) alloys. The highest magnetic moment was found for the $Mn_{50}Al_{46}Ni_4$ alloy. Total energy calculations point to a more stable antiferromagnetic configuration of the $Mn^{1a}-Mn^{2e}$ pair. DTA measurements pointed out the formation of the metastable τ phase around 470 °C and its decomposition into the stable γ_2 and β -Mn(Al) phases at 850 °C. The τ phase was found along with the ε' phase only in the as-cast sample and the ones annealed at 470° °C. A maximum τ phase concentration of 50% was found for the sample annealed at 470 °C for 6 h. The close values of the theoretical and experimental effective magnetic moments confirm the existence of the ε' and τ phases in these alloys.

Experimental and Computational Details:

- Electronic structure calculations were performed in the framework of the Local Density Approximation (LDA) of the Density Functional Theory by means of the SPR-KKR method using the experimentally determined lattice parameters for $Mn_{54}Al_{46}$ ($a = 3.94$ Å, $c = 3.58$ Å). All relativistic effects have been taken into account, including the spin-orbit coupling, in the ferromagnetic (FM) and antiferromagnetic (AFM) spin configurations of the $Mn^{1a}-Mn^{2e}$ pair.
- The $Mn_{54}Al_{46}$ and $Mn_{50}Al_{46}Ni_4$ ingots were prepared by induction melting of the starting components under a purified Ar atmosphere.
- The samples were annealed in an inert Ar atmosphere at temperatures between 470 and 850 °C for different times followed by quenching in water.
- XRD investigations were performed using a Brüker D8 Advance X-ray diffractometer with Cu K_α radiation.
- Differential thermal analysis (DTA) was performed between 100 and 900 °C under Ar atmosphere with a temperature ramp rate of 20 °C/min.
- The magnetization and magnetic susceptibility were measured with a Weiss-type magnetic balance in a temperature range of 300-800 K.

Electronic Structure Calculation Results:

Mn-Al system \rightarrow **Mn and Al atoms are situated in alternating planes** spaced at a distance of $c/2$ [1].

Mn atoms situated in **adjacent planes** ($Mn^{1a}-Mn^{2e}$ pairs) \rightarrow **AFM coupling**.

In order to **decrease** the weight of the **AFM interactions** \rightarrow **Ni or Ti substitutions** for Mn in $Mn_{54}Al_{46}$.

Smaller Ni or Ti moments \rightarrow **increase** of the **total magnetic moment** for $Mn_{50}Al_{46}M_4$ ($M = Ni, Ti$).

➤ **Smaller total energy** for a **mixed 1a and 2e Ni occupancy** than for an exclusive 2e Ni occupancy.

➤ **A lower total energy** was found for the alloy with the **AFM coupled $Mn^{1a}-Mn^{2e}$ pair**.

Calculated magnetic moments for the τ phase of $Mn_{50}Al_{46}M_4$ ($M = Ni, Ti$), $Mn_{50}Al_{50}$ and $Mn_{54}Al_{46}$ alloys.

| | | Mn | | Al | | M | | Total | |
|---|------------------|--------------|--------------|--------------|--------------|--------------|--------------|--------------|--------------|
| | | $m_s(\mu_B)$ | $m_l(\mu_B)$ | $m_s(\mu_B)$ | $m_l(\mu_B)$ | $m_s(\mu_B)$ | $m_l(\mu_B)$ | $m_s(\mu_B)$ | $m_l(\mu_B)$ |
| Mn ₅₀ Al ₅₀ | | 2.41 | 0.04 | -0.08 | - | - | - | 4.68 | 0.08 |
| Mn ₅₄ Al ₄₆ (FM) | Mn ^{2e} | 2.43 | 0.007 | -0.08 | - | - | - | 5.02 | 0.07 |
| | Mn ^{1a} | 2.39 | 0.04 | | | | | | |
| Mn ₅₄ Al ₄₆ (AFM) | Mn ^{2e} | -3.17 | -0.01 | -0.07 | - | - | - | 4.38 | 0.08 |
| | Mn ^{1a} | 2.35 | 0.04 | | | | | | |
| Mn ₅₀ Al ₄₆ Ti ₄ | | 2.34 | 0.04 | -0.08 | - | -0.66 | -0.004 | 4.43 | 0.07 |
| Mn ₅₀ Al ₄₆ Ni ₄ | | 2.45 | 0.04 | -0.08 | - | 0.63 | 0.06 | 4.85 | 0.08 |

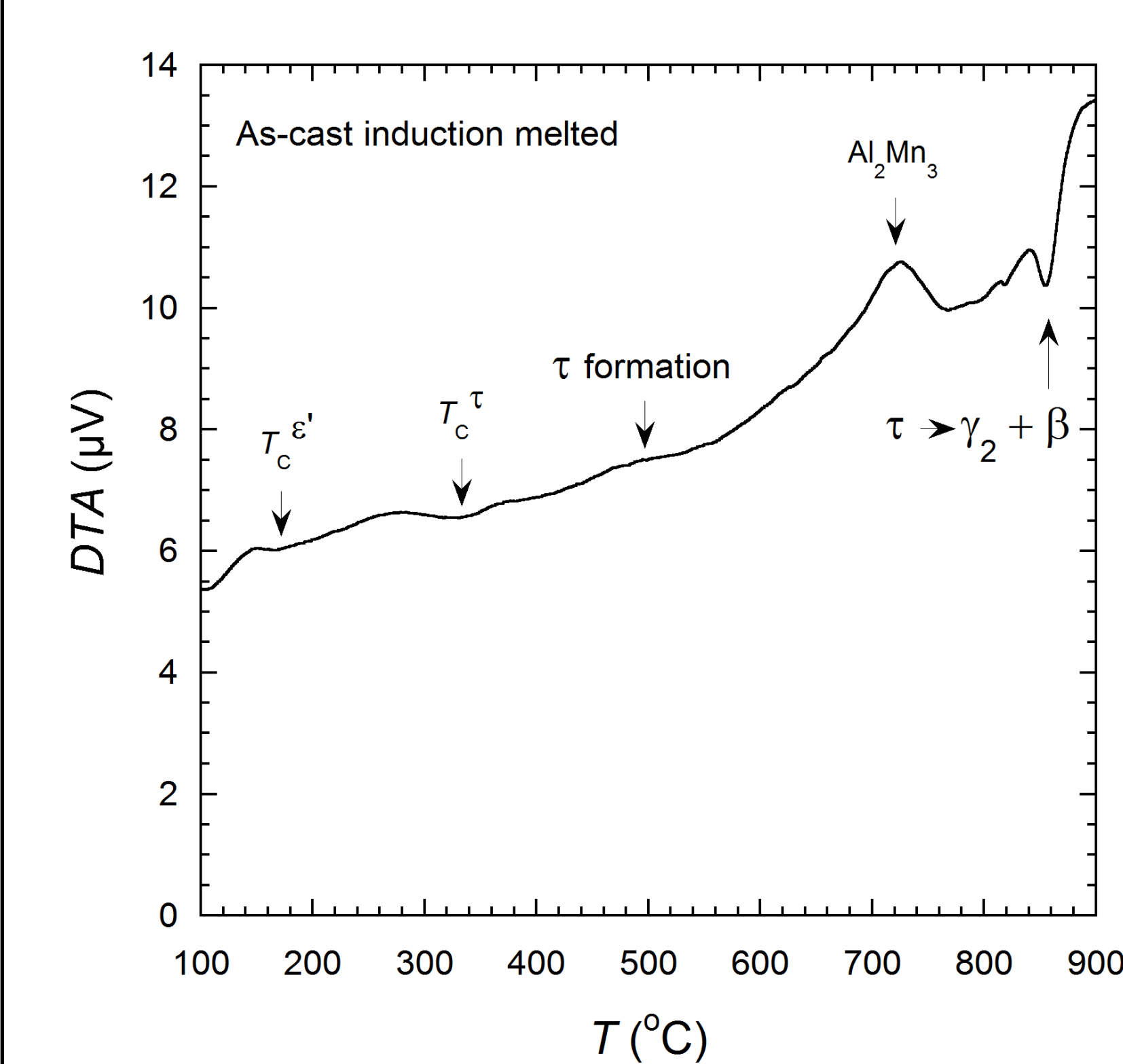
Experimental Results:

During the formation of the τ phase, the following transformation occurs [1,2]:

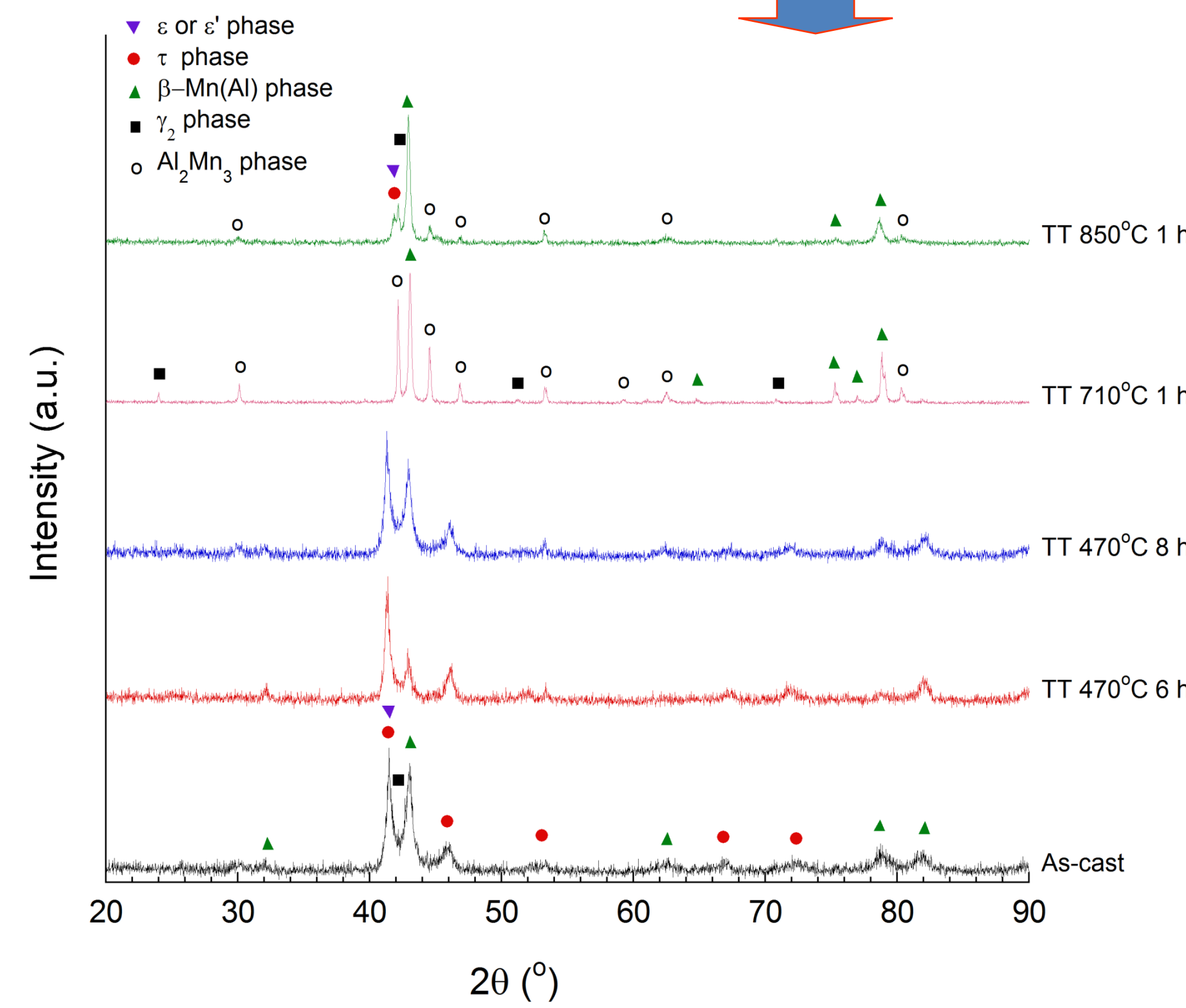
ε (disordered hexagonal) \rightarrow ε' (ordered orthorhombic) \rightarrow τ (tetragonal)

The ε' phase is an **ordered ε phase** [1,2] \rightarrow **XRD peaks** of these two phases **superimpose**.

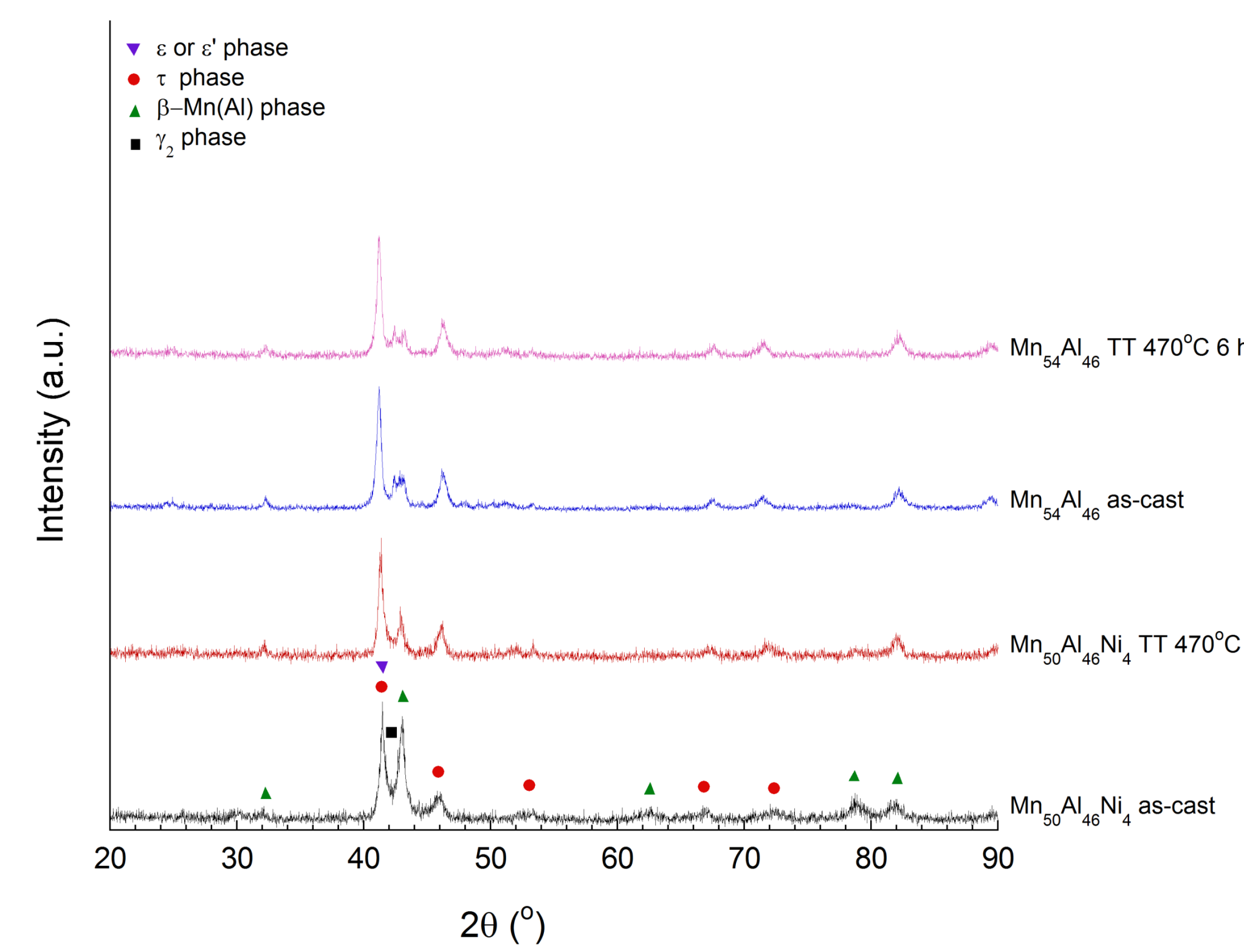
Magnetic measurements can **confirm** whether the phase is ε or ε' .



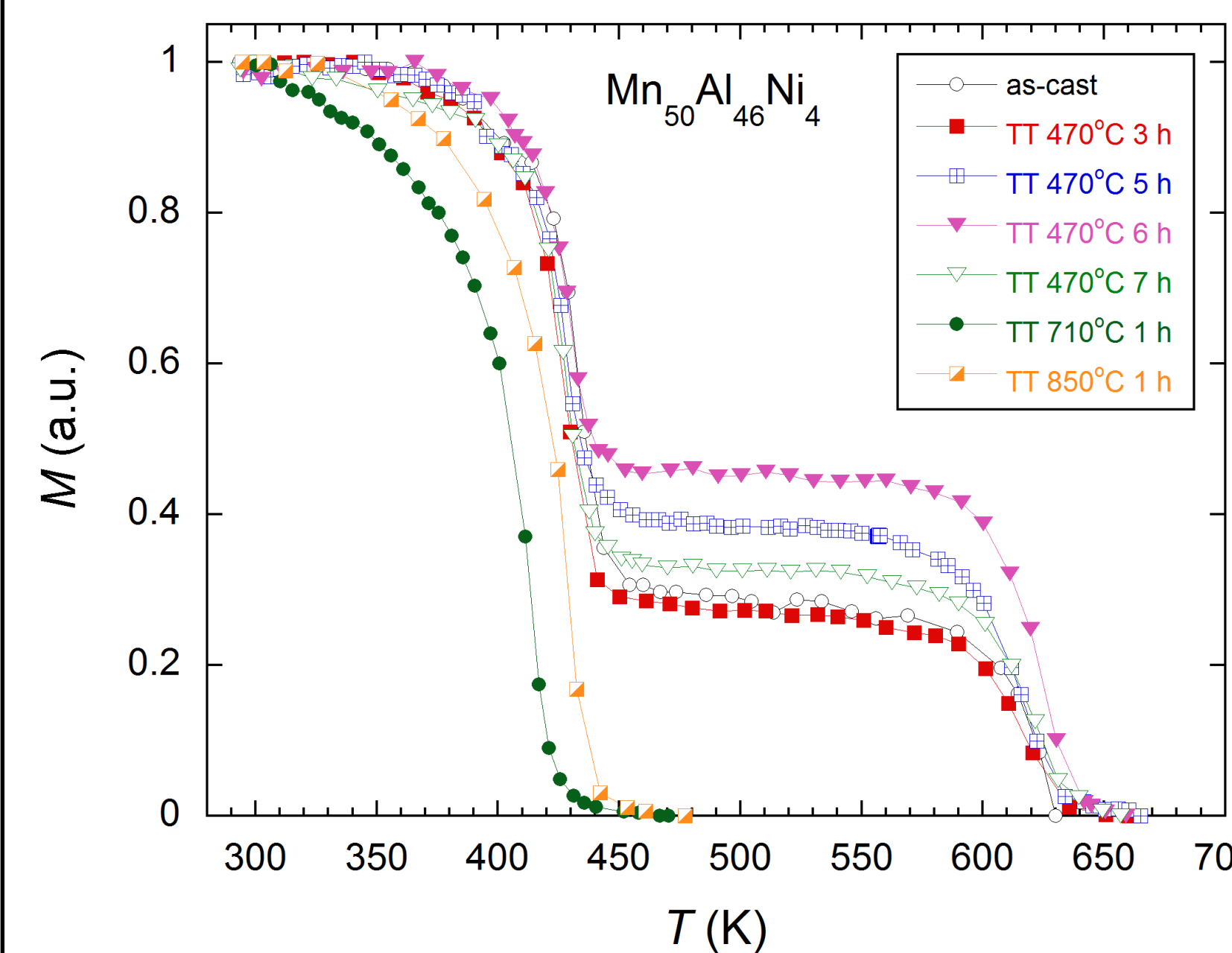
DTA curve for the as-cast $Mn_{50}Al_{46}Ni_4$ alloy.



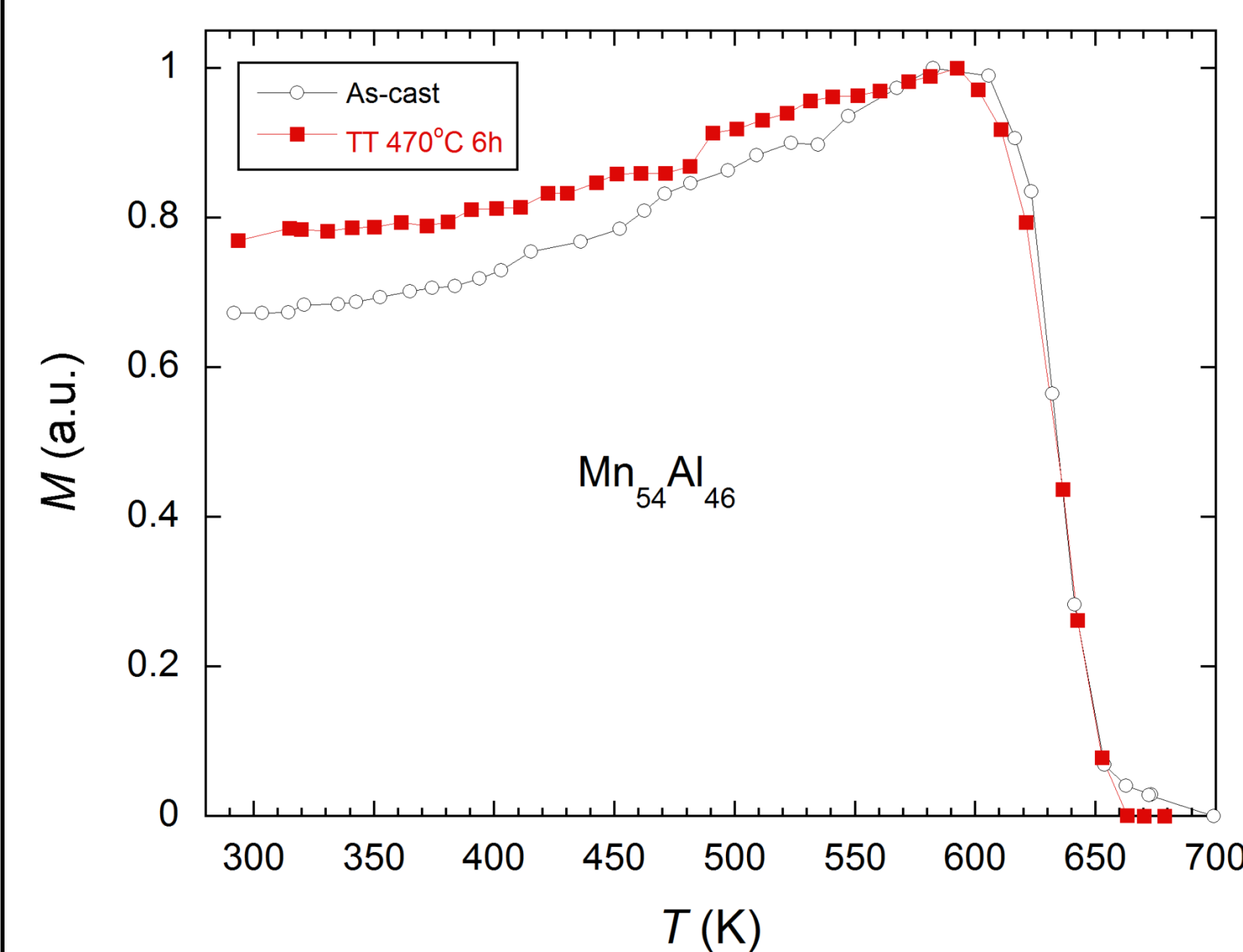
XRD patterns for the $Mn_{50}Al_{46}Ni_4$ alloy after annealing at different temperatures.



XRD patterns for the as-cast and annealed $Mn_{54}Al_{46}$ and $Mn_{50}Al_{46}Ni_4$ alloys.



Temperature dependence of the normalized magnetization of the $Mn_{50}Al_{46}Ni_4$ alloy after annealing at different temperatures.



Temperature dependence of the normalized magnetization of the as-cast $Mn_{54}Al_{46}$ alloy and the sample annealed at 470° °C for 6 h.

As-cast + TT 470 °C 6h

Two magnetic phases:
 τ (high T_C) and ε' (low T_C).

TT 710 °C + 850 °C

single magnetic phase behavior (lower T_C)

$Mn_{54}Al_{46}$ samples

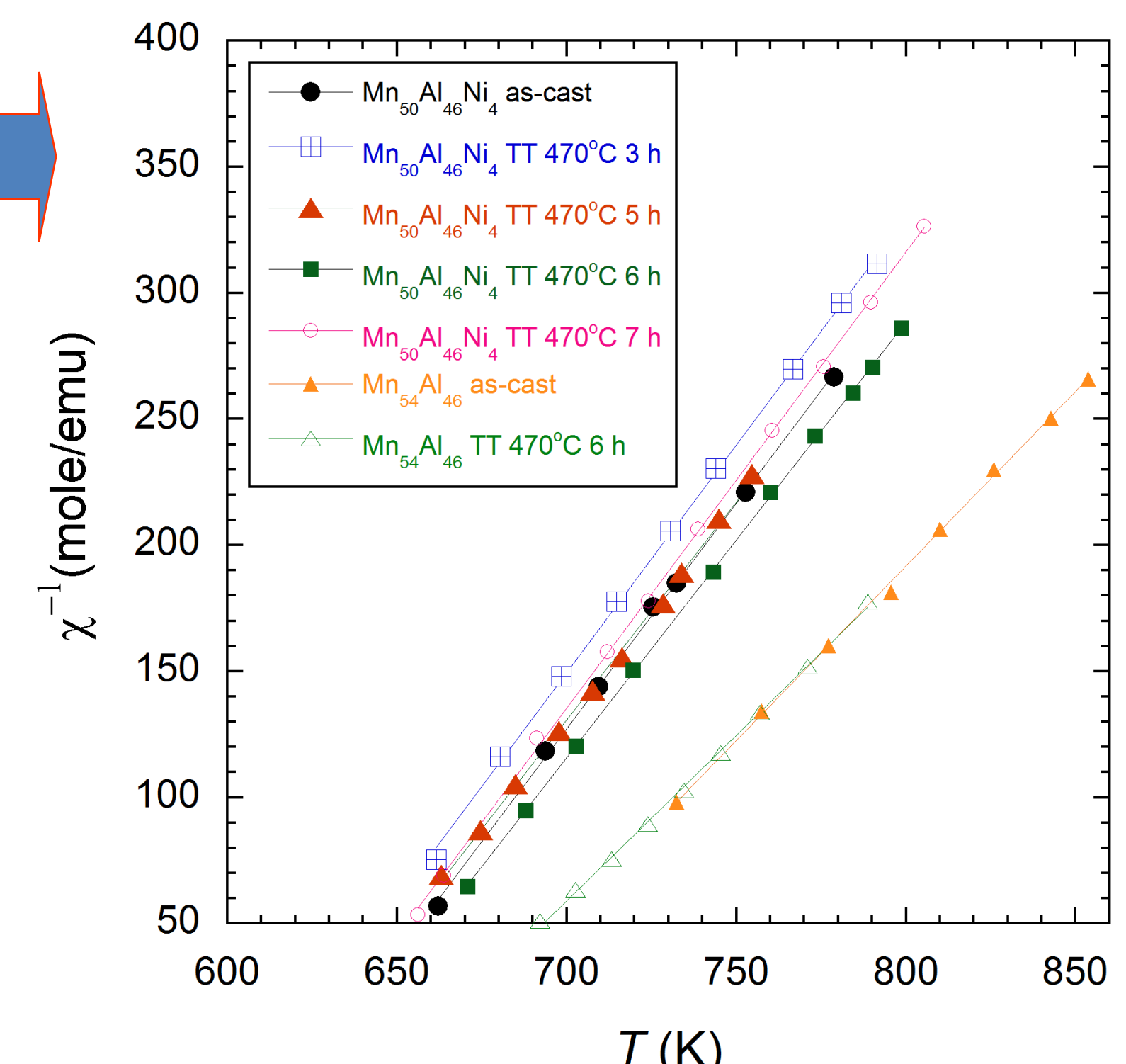
single magnetic phase behavior (τ phase).

The experimental data fits the Curie-Weiss law $\chi = C/(T-\theta)$ with the parameters given in the table below.

The average experimental Mn moment in the ordered state was calculated taking into account the concentrations $c_{\varepsilon'}$ and c_{τ} of the ε' and τ phases and the theoretical values of Mn moments in the two phases ($\mu_{\varepsilon'} = 1.88 \mu_B/\text{Mn atom}$ and $\mu_{\tau} = 2.37 \mu_B/\text{Mn atom}$ [1]) according to the relation: $\mu_{\text{exp}} = c_{\varepsilon'} \mu_{\varepsilon'} + c_{\tau} \mu_{\tau}$.

The μ_{eff} values, the Mn spins S , the τ phase content c_{τ} , the average Mn moments in the ordered state μ_{teor} and μ_{exp} , the paramagnetic Curie temperatures θ , and the Curie temperatures T_C of the τ phase of the $Mn_{50}Al_{46}Ni_4$ and $Mn_{54}Al_{46}$ alloys.

| Sample | $\mu_{\text{eff}}/f.u.$ (μ_B) | μ_{eff}/Mn (μ_B) | S_{Mn} | c_{τ} (%) | μ_{teor} (μ_B) | μ_{exp} (μ_B) | θ (K) | T_C (K) |
|------------------------------------|--|--------------------------------------|----------|-------------------|------------------------------------|-----------------------------------|-----------------|--------------|
| $Mn_{50}Al_{46}Ni_4$ as-cast | 2.11 | 2.98 | 1.07 | 35 | 2.14 | 2.05 | 629 | 624 |
| $Mn_{50}Al_{46}Ni_4$ TT 470 °C 3 h | 2.10 | 2.97 | 1.07 | 33 | 2.14 | 2.04 | 617 | 625 |
| $Mn_{50}Al_{46}Ni_4$ TT 470 °C 5 h | 2.15 | 3.04 | 1.1 | 43 | 2.2 | 2.09 | 624 | 635 |
| $Mn_{50}Al_{46}Ni_4$ TT 470 °C 6 h | 2.15 | 3.04 | 1.1 | 50 | 2.2 | 2.13 | 633 | 632 |
| $Mn_{50}Al_{46}Ni_4$ TT 470 °C 7 h | 2.10 | 2.97 | 1.07 | 38 | 2.14 | 2.07 | 625 | 629 |
| $Mn_{54}Al_{46}$ as-cast | 2.40 | 3.27 | 1.21 | 100 | 2.42 | 2.37 | 662 | 643 |
| $Mn_{54}Al_{46}$ TT 470 °C 6 h | 2.47 | 3.36 | 1.25 | 100 | 2.50 | 2.37 | 655 | 644 |



Reciprocal susceptibility versus temperature curves of the $Mn_{50}Al_{46}Ni_4$ and $Mn_{54}Al_{46}$ alloys. The lines represent the Curie-Weiss fit of the experimental data.

Close values of the μ_{teor} and μ_{exp} values **confirm** the existence of the **ε' and τ phases** in the investigated $Mn_{50}Al_{46}Ni_4$ alloy.

Conclusions:

- The **largest calculated magnetic moment** was found for the **$Mn_{50}Al_{46}Ni_4$** alloy.
- A **lower total energy** was found for the alloy with the **AFM coupled $Mn^{1a}-Mn^{2e}$ pair**.
- All of the samples are phase mixtures and are ferromagnetic at room temperature. The **τ phase was found along with the ε' phase** only in the **as-cast $Mn_{50}Al_{46}Ni_4$ sample** and the ones **annealed at 470 °C** with a **maximum τ phase concentration of 50%**.
- **Close values of the μ_{teor} and μ_{exp} values confirm the existence of the ε' and τ phases** in the investigated $Mn_{50}Al_{46}Ni_4$ alloy.
- Ongoing experimental work is currently being conducted to obtain the τ phase as a majority phase in these compounds.